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Nano-sized Laser Beams without Diffraction Spreading

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ABSTRACT

Whereas exciting progress has been made to beat diffraction in optical spectroscopy [1-8], making a nano-sized laser beam remains challenging due to diffraction spread [9,10]. Using Bethe's expression for the optical transmission coefficient of a circular hole in a perfect conductor screen of zero thickness [9,11]. T=1024 $\pi^3 a^6/27\lambda^4$, we found that the transmission of light with a wavelength λ =800 nm through such a circular hole of a radius of 1 nm is about ~2.3 × 10⁻⁶. So a decent nanosized laser beam for miniaturization of optical elements is not available. Here we show that coupling Surface Plasmon-Polaritons (SPPs) to appropriate dielectric material can result in fundamentally diffraction-free down-sized, especially nano-sized laser beams. For example, the composite structure Si₃N₄/Au(44.5nm)/SiO₂ (456nm)/(SiO₂, Si₃N₄, SiO₂) can achieve a nano-sized laser beam of about half the incoming light intensity. This approach, by transforming the macroscopic laser beams into multiple nano-sized laser beams, holds promise for ultrafast laser imprinting of nanopores for DNA sequencing and other miniature photonic devices in optical signal processing industries.

KEYWORDS

Optical spectroscopy, Nano-sized Laser Beams, DNA sequencing

INTRODUCTION

In optics, diffraction renders a blurred Airy disk image for a point object and thus creates a resolution limit to an imaging instrument [1]. Over the past decades, tremendous progress has been made to improve the resolutions of optical images. Hell et al. used a depletion laser beam to squeeze the non-depleted excitation fluorescent spot to a tiny point and achieved a 35 nm resolution of molecule images [2,3]. Zhuang et al. reconstructed 20 nm-resolution images of individual photo-switchable fluorescent molecules that are turned on and off by excitation light of different colors [12]. Betzig et al. used photo-activated localization microscopy to image intracellular proteins at a nanometer resolution [5]. Pendry proposed to use survived evanesce waves in negative-index media to make a perfect image [6], which was realized experimentally by Zhang et al., [7]. Yablonovitch et al. developed a 3D tapered metal-insulator-metal nano-gaped device that delivers background-free near-field images with a deep sub-wavelength resolution.

While these landmark advances are impressive in beating diffraction in image optics, overcoming the diffraction spreading of a laser beam is rarely studied. Light from a sub-wavelength aperture usually diffracts in all directions, which makes a nano-sized laser beam virtually impossible [9]. Townes et al. first discussed the possible suppression of rapid diffraction- spreading of a laser beam in nonlinear dielectric media by an intense laser [10]. Lezec et al. created an almost-directional light beam out of metallic films of a bull's eye structure, whose practical application is limited by the complicated fabrication of periodic grooves [13]. Noticing the importance of beating diffraction in science [1-8], here we propose an approach to make a nano- ized laser beam for technological applications

TRANSFER MATRIX MODELLING

We use a planar metal-dielectric tri-layer film to absorb laser light as SPPs at a conductor/dielectric interface [14,15] and add another top dielectric layer to reduce SPPs to free- space optical waves. This conversion does allow photons traveling directionally without diffraction [16].

We calculate the intensity of light reflected by and transmitted through this coupling structure [17-21] with a transfer-matrix method. The transfer-matrix method is briefly summarized as follows. Suppose the coupling structures are composed of optically isotropic and



homogeneous N planar layers. Each layer i has a thickness di. Both the width of the first layer and that of the last layer are assumed to be infinite. The dielectric properties of each layer i are characterized by its complex dielectric functions $\epsilon_i(\omega)$.

We use the dielectric functions calculated from the refractive index of materials (https://www.filmetrics.com/refractive-index-database). A p-polarized monochromatic plane wave with wavelength λ_0 is incident on the stack from layer 1 through layer N, making an arbitrary angle of incidence θ_1 with respect to the surface normal of the first interface. We describe the electric field in layer *i* as a superposition of the reflected and transmitted plane waves. For example, for a single ideal interface at the O-xy plane, the electric field below and above this interface is written as $\mathbf{E}(x, z < 0) = \mathbf{E}e^{i}(k\mathbf{1}_{x}x + k\mathbf{1}_{z}z - wt) + \mathbf{E}'ei(k\mathbf{1}_{x}x - k\mathbf{1}_{z}z - wt)$ and $E(x,z>0) = \mathbf{E}e^i(k2_xx+k2_zz-wt) + \mathbf{E}'e^i(k2_xx-k2_zz-wt)$. Here the wavevector for the incident plane wave is $k_{1x}\,\mathbf{i}$ + $k_{1z}\,\mathbf{k}$, the wave-vector for the reflected plane wave is $k_{1x}\,\mathbf{i}$ + $k_{1z}\,\mathbf{k}$ in layer 1, and so on for the wave-vectors in layer 2. The x-components of all the wave-vectors are equal, i.e., $k_{1x} = k_{2x} = ...k_{Nx} = 2\pi n_1 \sin\theta_1 / \lambda_\circ$. The z-components of the wave-vectors can be written as $k_{iz} = 2\pi \sqrt{\varepsilon_i - \varepsilon_1 \sin^2 \theta_1 / \lambda_o}$. The transformation of the incident and reflected fields across each interface *i* of two neighboring *i*-th and (*i*+1)-th layers are related to each other by an interface transfer matrix T(i), i.e.,

$$\begin{bmatrix} E_{(i+1)x} \\ E'_{(i+1)x} \end{bmatrix} = T(i) \begin{bmatrix} E_{ix} \\ E'_{ix} \end{bmatrix} \text{ with}$$

$$T(i) = \frac{1}{2} \begin{bmatrix} 1 + \frac{\varepsilon_i k_{(i+1)z}}{\varepsilon_{(i+1)} k_{iz}} & 1 - \frac{\varepsilon_i k_{(i+1)z}}{\varepsilon_{(i+1)} k_{iz}} \\ 1 - \frac{\varepsilon_i k_{(i+1)z}}{\varepsilon_{(i+1)} k_{iz}} & 1 + \frac{\varepsilon_i k_{(i+1)z}}{\varepsilon_{(i+1)} k_{iz}} \end{bmatrix}$$

$$(1)$$

Similarly, when the optical fields travel in a single j-th layer of thickness dj, the field amplitudes in the upper- and down-sides of this layer are related by a propagation transfer matrix T(dj), i.e.

$$\begin{bmatrix} E_{jx}(z_{j} = d_{j} + z_{j-1}) \\ E'_{jx}(z_{j} = d_{j} + z_{j-1}) \end{bmatrix} = T(d_{j}) \begin{bmatrix} E_{jx}(z_{j} = z_{j-1}) \\ E'_{jx}(z_{j} = z_{j-1}) \end{bmatrix}, \text{ with}$$
$$T(d_{j}) = \begin{bmatrix} e^{ik_{2z}d_{j}} & 0 \\ 0 & e^{-ik_{2z}d_{j}} \end{bmatrix} e^{ik_{zx}d_{j} tg\theta_{2}}$$
(2)

For a multilayer film, the electric fields at the first interface are related to those at the last interface by an overall T-matrix, which is the successive matrix products of the propagation transfer matrix and interface transfer matrix, i.e.,

$$\begin{bmatrix} E_{N-1x}(z_{N-1} = d_{N-1} + z_{N-2}) \\ E'_{N-1x}(z_{N-1} = d_{N-1} + z_{N-2}) \end{bmatrix} = T_{total} \begin{bmatrix} E_{1x}(z_1 = 0) \\ E'_{1x}(z_1 = 0) \end{bmatrix},$$

Where,
 $T_{total} = T(N-1)T(d_{N-1})\cdots T(3)T(d_3)T(2)T(d_2)T(1).$ (3)

If we set the incident and reflected electric field from the bottommost layer as one and r = E'1 / E1 and the transmitted and reflected field from the top-most layer as t = EN/E1 and 0, the matrix relation can then be expressed as:

$$\begin{bmatrix} t\cos\theta_N\\0\end{bmatrix} = \begin{bmatrix} T_{11} & T_{12}\\T_{21} & T_{22}\end{bmatrix}\begin{bmatrix} \cos\theta_1\\r\cos\theta_1\end{bmatrix}$$
(4)

Thus the transmission and reflection coefficients of the final layer are given by

$$t = (T_{11} - T_{12}T_{21} / T_{22})\cos\theta_1 / \cos\theta_N$$
(5)
and (5)

$$r = -T_{21} / T_{22} \tag{6}$$

Finally, the reflectance R, the transmittance T, and the absorbance A are then given by R = \mid r² \mid ,

$$T = \operatorname{Re}\left\{\frac{\mathbf{n}_{N}\cos\theta_{N}}{\mathbf{n}_{i}\cos\theta_{i}}\right\}\left|t\right|^{2} \text{ and } A = 1 - R - T$$

REFLECTANCE AND TRANSMITTANCE OF METAL-DIELECTRIC FILMS

We use this transfer matrix method to design our material structure. At first, we use a tri- layered film $Si_3N_4/Au/SiO_2$ stack to absorb laser light into SPPs with Kretschmann prism coupling configuration [22]. As shown in Figure 1, the 800 nm wavelength laser light with an angle of incidence is 49.1° is completely trapped at the interface of this film if the thickness of the smooth gold film [23] is about 44.5 nm.



Figure 1: The reflectance, transmittance, and absorbance of the film $Si_3N_4/Au(44.5nm)/SiO_2$ stack vs the angle of incidence. The p-polarized monochromatic light with wavelength 800 nm casts on the film. The surface Plasmon resonance angle is found to be θ_{spp} =49.1°.

We then put another Si₃N₄ layer onto the Si₃N₄/Au(44.5nm)/SiO₂ film with varying SiO₂ thickness. As shown in Figure 2, the transmittance of the film Si₃N₄/Au(44.5nm)/SiO₂/Si₃N₄ is 0~45.8% when the same light incident on it with the same angle of incidence. SPPs can couple efficiently with Si₃N₄ with a maximum transmittance of 45.8% in the sample Si₃N₄/Au(44.5nm)/SiO₂(456nm)/Si₃N₄.



Figure 2: The reflectance, transmittance, and absorbance of Si₃N₄/Au(44.5nm)/SiO₂/Si₃N₄ film vs SiO₂ thickness. When p-polarized wavelength 800 nm light with casts on the sample Si₃N₄/Au(44.5nm)/SiO₂(456nm)/Si₃N₄ with an angle of incidence is θ_{spp} =49.1°, the maximum transmittance is about 45.8%.

To understand the physics for this optical transmission, we also



calculate the magnetic field in the i-th layer of the sample Si_3N_4 / Au(44.5nm)/SiO₂(456nm)/Si₃N₄, which takes the following form. $H = \omega \varepsilon (E - E')/k \qquad (9)$

$$H_{yi} = \omega \varepsilon_i (E_{ix} - E'_{ix}) / k_{iz}$$

Here the x-components of the electric fields, including both the incident electric fields and the reflected electric fields in each layer, are also calculated by the transfer matrix formulation (3-6). As shown in Figure 3, the magnetic field amplitude reaches a maximum at the Au/SiO₂ interface and decays exponentially in the SiO₂ layer as plasma-polaritons. When entering the SiO_2/Si_3N_4 interface, the magnetic field stops declining and starts to travel through the Si₂N₄ layer as an un- attenuated oscillation. The SiO₂/ Si₂N₄ interface acts as an excitation source for this un- attenuated oscillation. This interface behaves just like shaking one end of a rope to generate a transverse string wave in the entire rope. This emission of light is selfdirectional, i.e., is diffraction-free, and is quite different from surface plasmon-coupled emission [24], or direction- selective emission in the metal-dielectric-metal structure [25] or light-beaming caused by the interaction of SPPs with surface grooves in the metal [13], or perfect transmission of light through pinholes in a Fabry-Perot cavity [26]. The top dielectric layer acts as a vent for SPP photons to emit, and the more surface area of the dielectric, the more free photons one can get.



Figure 3: The magnetic field Hy in the sample Si_3N_4 /Au(44.5nm)/ SiO₂(456nm)/Si₃N₄ vs. sample depth. The magnetic field amplitude reaches the maximum at the Au/SiO₂ interface and then decays exponentially in the SiO₂ layer to form plasma polarities. When entering the SiO₂/Si₃N₄ interface, the magnetic field in the SiO₂₁ stops declining and starts to travel through the Si₃N₄ layer as un-attenuated oscillations.

This dielectric/metal/dielectric/dielectric structure can modify the wave-fronts of electromagnetic waves by the top dielectric material to turn laser light on and off simultaneously. For example, composite structure Si₃N₄/Au(44.5nm)/ SiO₂(456nm)/ (SiO₂, Si₃N₄, SiO_2) is obtained by sideway connecting the SPP structure $Si_3N_4/$ Au(44.5nm)/ SiO₂ with the coupling structure Si_3N_4 /Au(44.5nm)/ $SiO_{2}(456nm)/Si_{2}N_{4}$, as shown in Figure 4. This composite structure can trap most SPP photons at the gold-silicon interface, and a certain amount of SPPs becomes propagating photons in the capping layer Si₂N₄. Since SPPs scatter along with the interface when entering the lateral Si₂N₄/SiO₂ boundary, this SPP-scattering does not affect the far-field emission at all [27]. Therefore the beam size of this light transmission depends solely on the contact surface area of the top dielectric. One would expect the cross-section area of the resultant laser beam is the contact area of the topping dielectric material multiplied by a factor of $\sin\theta 1$. So this structure is expected to make an optical beam of a few nanometers with no diffraction spreading. For example, if the contact area of the coupling layer Si_2N_4 is a few nanometers square and the thickness of SPP hosting layer SiO, is 456 nm thick, the incident laser could become a nano-sized laser beam with a tremendous transmittance of 45.8%.



Figure 4: The light is partially turned on and off in the composite structure $Si_3N_4/Au(44.5nm)/SiO_2(456nm)/(SiO_2, Si_3N_4, SiO_2)$. The beam size of this dielectric-coupled SPP emission depends solely on the contact area of the top dielectric coupling layer Si_3N_4 , which turns the light on (hence it stores a bit "1")

CONCLUSIONS

In summary, we performed a systematic study of the optical properties of SPP nanostructures coupled with dielectric materials. While this study has not been experimentally realized, there is no theoretical obstacle to build a diffraction-free laser beam by coupling SPP nanostructures with appropriate dielectric material. The capping layer alignment shows significant flexibility in building structures of interest in a variety of applications. The generation of multiple beams from a single laser source could be used to design new optical storage for fast batch-bit access in big-data industries, or to develop new 3D-printing technology to fabricate nano-porous membranes or nanofibers for efficient industrial-scale hydrogen generation [28] and CO₂ electrolysis for global dioxide emission reduction [29-31]. The merit of this diffraction suppression is that a decent nano-sized laser beam could be made possible. A continuous effort aiming at improving the optical field [32,33] of this sub-wavelength laser beam for the fabrication of a solid-state nanopore for DNA sequencing [34-37] is also possible.

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